The Effect of Silicon CarbideAddition into Fibreglass Reinforced Composites

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Abstract In today's modern world the composite materials have been widely used not only for aerospace/aeronautics, but also automotive, sports and construction industries. The flexural strength of laminated composites depends on the characteristics of dispersive and matrix phases, considering the presence of tensile and compressive loadings. For this reason, the addition of particles into the matrix phase has been investigated to enhance its stiffness and consequently the elastic modulus of the composite. This work investigates a glassfibre composite reinforced with silicon carbide particles. The following responses were evaluated: apparent density, water absorption, flexural strength, modulus of elasticity. The results were compared to those manufactured with no silicon carbide particles. It was observed that the silicon carbide particles provided the increase of the apparent density and flexural strength of the composites.

Keywords Laminated Composites, Silicon Carbide, Fibre Glass, Hybrid Composites

1. Introduction

The growing demand of composite materials has increased the attention to the theories of reinforced materials and advanced processes for their production.

Some industrial sectors, as the areas of missiles, rockets and aircraft, owned the restricted use of structural polymeric materials or advanced polymeric materials. Following decades of restriction, modern industries have expanded the use of these materials. The automotive, sports and construction industries[1,2] have experienced an increase of this technology every year due to their superior specific strength, lower density, higher corrosion resistance and high strength/modulus over conventional materials[3,4].

Replacement of metals by composites in commercial aircraft prime structures has been rapidly increasing to achieve further structural weight saving and reduce fuel consumption with a background of globally soaring oil prices[3].

Brazil has expanded their experience of innovation in the application of structural composites, especially in the aeronautics sector, using for external and internal planes/ helicopters (about 20% of the area of an aircraft) and a lesser extent for structure of rockets[3].

The dynamic flexural loadings in aircraft fuselage must be considered in the structural design. In this case, the composites must provide high stiffness and fatigueresistance[5, 6].

The modification of the mechanical behavior of polymers by the addition of filler materials have shown a great importance in researches and has lately been a subject of considerable interest[7,8].

In order to increase the flexural modulus of laminated composites, ceramic particles have been added in the matrix phase to enhance the mechanical performance of laminated composites. Flexural strength of laminated composites can be improved with ceramic particles[9]. Silicon carbide (SiC) particles can improve thermal responses of polymers[10] and some mechanical responses[11] of laminates.

This work investigates the physic-mechanical properties of hybrid polymeric composites reinforced with glass fibre and silicon carbide particles. The fatigue resistance and thermal properties will be the scope of future investigations.

2. Materials and Methods

2.1. Epoxy Resin

The epoxy resin was supplied by M/s. Alpha Resiqualy Company (São Paulo, Brazil). The addition of 66.66% of resin (RQ-0100RF) with 33.34% of hardening agent (RQ-0164RF PLUS) was mixed to obtain the matrix phase.

2.2. Reinforcements Phases

The glassfibre fabric of 240 g/m^2 was used in this experiment. The silicon carbide particles were supplied by M/s. Saint Gobain Company (Barbacena, Brazil) with

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particle size between mesh 200-325.

2.3. Experimental Conditions

The glass fibre reinforced composites were investigated by the use of matrix phase non-modified and modified with silicon carbide particles.

2.4. Manufacture

The specimens were fabricated by hand-lay-up process[12] according to the recommendations of Vison and Sierakowski[13] (Fig. 1).

For this research, the silicon carbide particles were sieved in mesh 200 - 325 US-Ty ler (the particles retained on 325 US Tyler, were used).

By volume percent, the composite was shared in 30% of reinforced phase and 70% of matrix phase.

Three layers of glass fibre were cut into squares of 200 x 150 mm. Then, the mass of fibre was measured and the fibre volume (V_f) of composite was calculated according eqn. 1:

$$V_{\rm f} = m_{\rm f} / \rho_{\rm f} \tag{1}$$

Where m_f is the mass of fibre and $\rho_f (2,5 \text{ g/cm}^3)$ is the density of fibre. According eqn. 2, the volume of matrix (V_m) was obtained:

$$V_{\rm m} = (V_{\rm f} \times 70)/30$$
 (2)

After, by density of matrix ρ_m (1,16 g/cm³), the matrix mass (m_m) was obtained according eqn. 3:

$$m_{\rm m} = V_{\rm m} \times \rho_{\rm m} \tag{3}$$

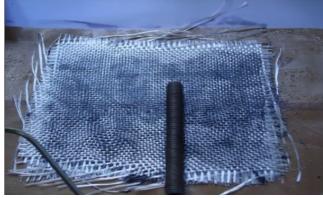


Figure 1. Hand lay-up method



Figure 2. Silicon Carbide addition

The weight of particles mass was based in 20 wt.% (this amount was based in results of[14]). The silicon carbide particles were mixed into epoxy resin matrix, manually.

Later, the matrix reinforced with particles, was used for laminating of the three layers of glass fibre.

The Armalon® demoulding sheets were used at the bottom and upper surfaces in order to ensure a goodsurface finish, as shown in Fig. 2.

The samples were cured under room temperature, under 20 Newtons of uniaxial compressive forces, for 7 days.

2.5. Methodology

2.5.1. Relative density

The apparent density was obtained according to ASTM SD792[15]. Test method a for testing solid plastics in water was used. By eqn. 4 was possible to calculate the relative density (rd) of sample:

$$rd = a/(a+w-b) \tag{4}$$

Where: a (apparent mass of specimen, without wire or sinker in air), b (apparent mass of completely immersed and of the wire partially immersed in liquid) and w (apparent mass of totally immersed and of partially immersed wire)[15].

2.5.2. Water Absorption

The water absorption was measured according to ASTM D570[16].

The specimens were immersed in saturated distilled water for 24 hours at 23°C. The water absorption was calculated as follows:

$$\% = \frac{wet \ weight - cond.weight}{cond.weight} \times 100 \tag{5}$$

Conditioned weight: is the mass of sample immersed in water.

The percentage of soluble matter lost during immersion, was determined by using the eqn.6.

$$\% = \frac{cond.weight - recond.weight}{cond.weight} \times 100$$
(6)

Recond. weight: mass of sample reconditioned in water after removal of there.

The percentage of water absorption was calculated dividing eqn.1 by eqn.2.

2.5.3. Flexural Stress

The samples were prepared according to ASTM D790[17], based on the recommendations of materials less than 1.6mm in thickness.

The flexural strength was calculated by Eqn. 7.

$$\sigma_f = 3PL/2bd^2 \tag{7}$$

Where P is the load at a given point of load-deflection curve, L (25,4 mm) is the support span, b (12,7 mm) is the width and d (mean of 1,28) is the depth of beam tested.

2.5.4. Modulus of Elasticity

The modulus was obtained by the tangent modulus of elasticity in the ASTM D790 as follows:

$$E_{\rm B} = (L^3 \text{ m})/(4\text{bd}^3)$$
 (8)

Where L (25,4 mm) is the support span, b (12,7 mm) is the width of beam tested, d (mean of 1,28 mm) is the depth of beam tested and m (calculated by graphic) is the slope of the tangent modulus (obtained by 50% deflection relative of the maximum flexural strength force).

3. Results

The comparisons between the results were carried out based on the Hypothesis Test for Difference of Means with Unknown Variances[18]. This analysis is able to verify whether the factor significant affects the response variables. The statistical analysis was performed for the flexural strength and modulus of elasticity responses. The other responses were analyzed based on the average and standard deviation of the experimental results.

Table 1 shows the means and standard deviations for each response variable.

 Table 1.
 Means and standard deviations

Variable Response	Without Particles		With particles	
	Mean	SD	Mean	SD
Relative density	1.56	0.096	1.63	0.005
Water absorption	5.83	0.056	2.69	0.008
Flexural stress	50.86	5.27	73.0806	9.63
Modulus Elasticity	6.65	0.77	6.63	0.44

Table 2 gives the amounts for the hypothesis test for two averages. The confidence of 95% was set, which corresponds to an α =level of 5%.

Table 2. Test: Hypothesis test						
V. R.	Нур.	t_{calc}	t _{tab}	Concl.		
Relative Density	$\begin{array}{l} H_{0}:\; \mu_{l} = \mu_{2} \\ H_{0}:\; \mu_{l} > \mu_{2} \end{array}$	6,52	1,81	Reject H ₀		
Flexural strength	$\begin{array}{l} H_0: \ \mu_1 = \mu_2 \\ H_0: \ \mu_1 > \mu_2 \end{array}$	4,52	1,81	Reject H ₀		
Flexural modulus	$\begin{array}{l} H_0: \ \mu_l = \mu_2 \\ H_0: \ \mu_l > \mu_2 \end{array}$	0,048	1,81	Accept H ₀		

The comparison was performed between two means based on the t-student distribution, calculating the value of the "t calculed" and comparing with "t tabulated". When the value of "t calculated" is outside the region of acceptance covered by the "t tabulated data", then the initial hypothesis (H_0) is rejected[18].

3.1. Relative Density

The apparent density values of the composites varied from 1.50 to 1.63g/cm³. Fig. 3 shows the density graph for both conditions. The addition of siliconcarbide particles leads to an increase of 4.15% in apparent density of the material. This behavior was expected due to the higher density of silicon carbide particles.

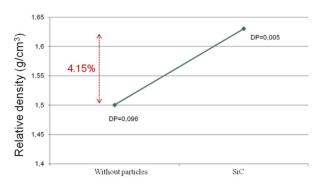
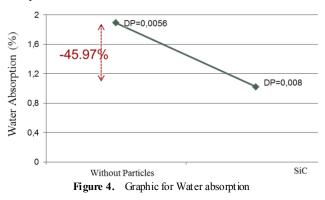


Figure 3. Graphic for Relative density

3.2. Water Absorption

The water absorption values of the composites decreased from 1.89% to 0.98%. Fig. 4 shows the water absorption graph for both conditions.

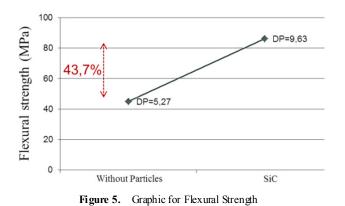
The silicon carbide addition provided the reduction of 45.97% on the water absorption of the composites. Like the research of[11], one barrier properties appears, by the high loading of silicon carbide, covering the fibres. This layer of particles under the fibres, promoted a decrease of water absorption.



3.3. Flexural Strength

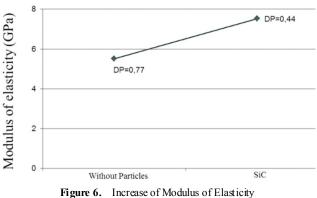
The flexural strength data of the composites varied from 50,86 to 73,08 MPa.

Based on the hypothesis test, the flexural strength was significantly affected by the addition of silicon carbide particles, exhibiting a rate of growing of 43,7%. Fig. 5 shows the increase in flexural strength.



The idea behind the increase is due the high silicon carbide particles density, then the particles were decanted (fig. 3) under glass fibre, then, according some arguments shown by[9] in his review, a fibre surface coating occurred. The particles promoted an increased in the roughening the fiber-matrix interface affecting the bonding strength between both, causing an increase in the shear stress transfer at the interface, collaborate to increase the flexural strength[9].

3.4. Modulus of Elasticity





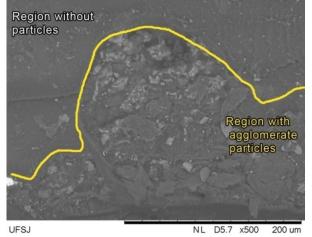


Figure 7. SEM - Agglomerates of silicon carbide particles in the inferior region of polymer

The modulus of elasticity varied from 6.65 to 6.63 GPa. The mean and standard deviation for each condition are shown in Table 1. Based on the Table 2, the two means are statistically the same (Table 2).

For this reason, it is concluded the particle addition did not affect this response. Fig. 6 shows the increase in modulus of elasticity.

The modulus did not affected by poor dispersion the particles into de epoxy matrix. Regions without particles in matrix phase can be observed (fig. 7). For improve the modulus, a good dipersion of particles, should be promoted.

4. Conclusions

The addition of micron silicon carbide particles provided the increase of apparent density and the reduction of water absorption of the composites. The addition of particles increased the flexural strength; however the modulus of elasticity was not significantly affected.

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